Polysulfone as a material for advanced electrical batteries

The materials based on polysulphone are used for polymer electrolite production. The rheological properties were measured for three component system polysulphone – solvent – propylene carbonate. The varying factor was propylene carbonate used as a plasticizer for polysulphone. Experiments were carried out in a wide shear rate and frequency ranges at 25 ºC. Polymer was characterized by the IR-spectral method. Rheological experiments were carried out using the technique of rotational rheometry. It was shown that the rheological behavior of all samples is linear and these three-component systems are viscoelastic liquids. It can be assumed that all systems under study at homogeneous in the absence of traces of water. The final goal of the study was the choice of the optimal composition (concentrations of a solvent and plasticizer) and the regime of preparing thin films from these systems which could provide obtaining solid state electrolyte for creating advanced electrical accumulator with a none-fragile polysulfone membrane. This part of the research is devoted to investigation of the rheological properties (viscosity and viscoelastic properties of composition of polysulfone – solvent – propylene carbonate systems.

Keywords: polysulfone, polymer solutions, plasticizer, viscosity, viscoelastic properties, solid-state electrolyte, thin-film membrane.

Introduction

Secondary sources of electrical energy are widely spread and used in the modern world. Nowadays, the basic element of such a kind is a lithium-ionic accumulator which is almost exclusively used in portable electronics as well as it is a perspective power supply for electric transport. Advantages of lithium-ionic accumulator are high store energy density, absence of a memory effect, and long term exploitation [1, 2]. Meanwhile, their essential defect is using liquid electrolytes. Thus does not allow for decreasing size of a power source to a film of several microns, impossibility to create flexible battery and provide necessary safety. Therefore, today we meet an increasing number of publications and research aimed to using a solid electrolyte/separator [3, 4].

The general way for building thin-film battery is creating a polymeric matrix which is acting of electrolyte and separator and this construction determines the mechanical properties of a battery in whole providing the formation of a stable 3D structure [5].

One of the most perspective polymers which can be used for these purposes is polysulfone. This term is common for the entire class of sulfo-group contacting compounds. A polymeric chain of these polymers contains para-aromatic groups bonded by esters and sulfonic group [6]. As a rule polysulfone is a rigid-chain amorphous polymer with relatively low molecular weight [7]. This is an engineering plastic with high mechanical performance in different modes of deformations (bending, extension, torsion) and high temperature stability (till 150–170 ± ºC). For this polymer, high chemical stability (in the pH range from 1 to 13), good resistance to oxidation [8]. In the same time this polymer can be dissolved in many aprotonic solvents such as dimethylacetamide and methylpyrrolidone. However there are serious difficulties in preparing polysulfone films because they appear brittle and do not resist to deformations perpendicular to the film plate.

All these reasoning were the cause for this study. It was proposed to plasticizer polysulfone by using propylene carbonate as a plasticizer. It was presumed that this liquid will occupy free volume in the polymer bulk when the solvent from the polysulfone solution will be completely deleted.

This part of the research work is devoted to investigation of the rheological properties (viscosity and viscoelastic properties of polysulfone – solvent – propylene carbonate systems.
Experimental

Objects. A sample of granulated polysulfone sample (produced under working conditions using the faculty of the Institute of Plastics (Moscow), PSF-150 grade according to the manufacturer’s standard) was used. N-Methylpyrrolidone purchased from LLC Ruskhim was used as the solvent. Solvent contained 0.1% of water (determined by the Fischer method). Propylene carbonate was bought from Chemical Ltd. The mass part of water was less than 0.05% (determined by the Fischer method).

Methods. Chemical structure of the polymer was confirmed by the Fourier IR spectroscopy using the IFS-66 v/s instrument (Bruker, Germany) — scan-30, resolution of 2 sm⁻¹, range of 0–4000 sm⁻¹.

Three component solutions were obtained in two stages. At first stage, a plasticizer was added to a given quantity of a solvent. The mixture was placed into a ultrasonic bath for 10 min. At the second stage, the polymer was introduced into the beforehand mixture and a sample stirred in a mechanical stirrer during 7 hours at a speed of 800 RPM under permanent heating at 100 °C. Samples with constant-polymer – solvent ratio was used and the quantity of plasticizer varied from 10 to 60% in relation to the polymer mass.

Rheological properties of all highly concentrated systems were measured on a rotational rheometer Kinexus Pro (MalvernPananalytical). Temperature was regulated using the Peltier element with thermal stabilization of the measuring system with an active cover. The latter allowed us to prevent heat lose and minimize temperature gradients inside a sample.

The procedure of measuring has been carried out using the cone-and-plate and plate-and-plate working units. The angle between the cone and the plate was 1°; the cone diameter was 40 mm. When the plate-and-plate system was used, the diameter of the user plate was 20 mm. All experimental data were obtained at 25 °C. The range of the given shear rate in measuring the apparent viscosity was from 0.01 to 100 s⁻¹. The number of points in scanning was 10.

In the amplitude test at measuring viscoelastic parameters of the samples, the amplitude of strain was varied from 0.01 % to 30 % at the frequency of 1 Hz.

In measuring frequency dependencies of the components of the dynamic modulus, the frequency varied from 0.01 to 50 Hz.

Results and Discussion

Figure 1 shows the IR spectrum of the used polysulfone sample. The aim of this test was to determine the exact molecular structure of the used polymer from great variety of possible polysulfones. As is seen, the obtained spectrum demonstrates the presence of n-substituted phenyl rings (690–850 cm⁻¹ band) corresponding to non-plane stretching vibrations C=C–H group, vibrations of C=C groups (1489 and 1585 cm⁻¹ bands), and C–H stretching vibrations (3060 cm⁻¹ band). 1000–1050 cm⁻¹ band corresponds to isopropylidene group, 1360–1420 cm⁻¹ band to C–C group, 2770–2840 cm⁻¹ to methyl groups, bands in the range 1150 and 1330 cm⁻¹ are responsible to sulfone groups, bands of 1242 и 1103 cm⁻¹ are related to Ph–O–Ph bonds and angles.

![Figure 1. IR-spectrum of polysulfone sample used in the work](image-url)
So, the analysis of the IR-spectrum and comparison with the database shows that our sample is identical to \( p \)-phenylsulfone.

The main problem in preparing polysulfone thin films is inherently connected with the pre-history of their solvation. Therefore just the rheology of polymer solutions can give answers and prompt the optimal protocol for obtaining films with necessary characteristics because the structure of solid polymer in a film is created in a previous solution.

Figure 2 shows the dependencies of the apparent viscosity on shear rate for a series of plasticized solutions. The viscosity of the initial sample (without a plasticizer) is 40 \( \text{Pa}\cdot\text{s} \). This solution demonstrates the Newtonian type of the rheological behavior. The same remains true for all plasticizer solutions.

![Figure 2. Dependencies of the apparent viscosity on shear rate for samples with different concentration of the plasticizer.](image)

So, there is a three-component system. This (Newtonian) character of the flow can be presumably, explained by a rather low concentration of the polymer in the composition. Indeed, the share of a plasticizer in the composition in whole varies from 3% to 15%. It corresponds to an increase of the total content of low-molecular weights liquids from 67% to 71%, i.e. it leads to dilution of a solvent. In this case, presumably, a system remains homogeneous and the decrease of the polymer concentration results to a decrease in the viscosity. Actually the viscosity of a system decreases as seen from Figure 2. This is also clearly seen in Figure 3 where the dependence of the viscosity of tree-components solutions on the plasticizer concentration is presented.

![Figure 3. Dependence of the viscosity of three-components systems on the concentration of the plasticizer.](image)
The results of measuring viscoelastic properties of the systems under study are presented in Figures 4 and 5.

Amplitude dependencies of the components of the complex elastic modulus are shown in Figure 4 for a single frequency of 1 Hz. One can see that the viscoelastic response remains linear in a rather wide amplitude range. It confirmed that the structure of all these systems is rather stable. In particularly it means that the results of measuring frequency dependencies of the elastic modulus are really relate to the linear domain of the viscoelastic behavior of solutions. These results are shown in Figure 5.

![Figure 4](image1.png)

Figure 4. Amplitude test for the storage modulus (a) and the loss modulus (b) made mod different concentration of the plastisizer (f = 1 Hz)

![Figure 5](image2.png)

Figure 5. Frequency dependencies of the component of the dynamic modulus: the storage modulus (a) and the loss modulus (b) for systems with different concentration of the plasticizer

For all compositions, experimental data show that $G'' \gg G'$, i.e. liquid component dominates and all compositions are elastic fluids. The slope of the storage modulus is 1.54 that is much lower than the value corresponding to the Maxwell singly-relaxation-time model. So, it is necessary to assume that relaxation spectra of solutions are rather wide. This is related to the presence of polymeric chains with their inherent segmental movement because polysulfone macromolecules are rather semi-rigid than rigid.

Experimental curves are placed rather systematically with exclusion of the most concentrated (50 %) solution. It is reasonable to suspect that this is a limit of solubility of polysulfone. Similar results have been obtained for a two-component polysulfone solutions in our previous publication [9].

Conclusion

The results of the rheological study of complex tree-component solution (polysulfone, solvent, plasticizer) have been presented. These results include data on the apparent viscosity and viscoelastic properties in
a wide concentration range. These data are important for the choice of the optimal composition and regime of preparing thin-film solid membrane and solid electrolyte for advanced electrical accumulators.

References

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Заманауи электр аккумуляторлары ушін полисульфонды қолдану туралы

Полимерді электролиттер мен сепараторды жасау ушін полисульфон негізіндеғі материалдар қолданылады. Полиолді полисульфон негізінде үш компоненті жұқа ушін реологиялық сипаттамалар (қорінуші түткырлықтың қысық жалдамдықтарының қалыптағы, түткырсірпдің кешен модулінің амплитудалық және қолдану кезінде тұтқырлықтары) зерттелетін. Бастапқы жүйе N-метилпирролидондың полимердің өрісіндегі, пластификатордың өрісінде пропиленкарбонат қолданылады, жұқасы еритінділердің реологиялық касиеттерінің пластификатор кәсіпқаға айналдығына тәуелділігін зерттеді. Эксперименттер 25 ºC қатарында, құқымдастық және жаңа куралының қолдану үшін дайындады. Бұл және реологиялық кәсіпқағы қасиеттің ерітінділерінің жаңа куралының жылу жылжымдарына және температуралық амплитудалық қолдану кезінде реологиялық кәсіпқағы қасиеттерінің кеңейсі бұл тұтқырлықтардың көследі. Реологиялық кәсіпкерлер үшін электролиттер және электролит жылу жылжымдарына және температура амплитудалық қолдану кезінде реологиялық кәсіпкерлердің жаңа куралының кеңейсі болады.

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Полисульфон как материал для современных электрических аккумуляторов

Материалы на основе полисульфона используются как основа для создания полимерных электролитов и сепараторов. Реологические свойства (в зависимости важности вязкости от скорости сдвига, а также амплитудные и частотные зависимости компонента модуля вязкогоугольного комплекса) были измерены для трехкомпонентных систем на основе полисульфона. Исходная система — это раствор полимера в N-метильтирилолоне. В качестве пластикатора применен пропиленкарбонат, в работе изучена зависимость реологических свойств растворов от концентрации пластикатора. Эксперименты прово-
длялись в широком диапазоне скоростей в диапазоне скоростей и частот сдвига при 25 °C. Исходный полимер характеризовался методом ИК-спектроскопии. Реологические эксперименты проводились с использованием техники ротационной реометрии. Было показано, что зависимость вязкости от скорости сдвига для всех исследованных образцов является ньютоновской, и такие трехкомпонентные системы являются вязкоупругими жидкостями. Можно предположить, что все исследуемые системы однородны при отсутствии следов воды. Конечной целью исследования был выбор оптимального состава (концентрации растворителя и пластикатора) и режима приготовления тонких пленок из этих систем, которые могли бы обеспечить получение твердого электролита для создания усовершенствованного электрического аккумулятора с гибкой полисульфоновой мембраной, выступающей в роли электролита и сепаратора. Данная работа посвящена исследованию реологических свойств, таких как вязкость и вязкоупругие свойства системы «полисульфон – растворитель – пропиленкарбонат» и зависимости их от состава и структуры растворов.

Ключевые слова: полисульфон, полимерные растворы, пластикатор, вязкость, вязкоупругие свойства, твердотельный электролит, тонкопленочная мембрана.